

Fabrication and Characterization of Lanthanide-TiO₂ Nanotube Composites

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Abstract

Titanium dioxide (TiO₂) doped with neodymium (Nd) and/or Gadolinium (Gd) rare-earth elements were fabricated into nanotubes via the hydrothermal method in a KOH solution and *in-situ* doping. Titanium dioxide nanotubes (TNTs) and *in-situ* Nd-doped and/or Gd-doped TNTs were characterized with transmission and scanning electron microscopy, energy-dispersive X-ray analysis, X-ray diffraction, Raman spectroscopy, and Fourier-transform infrared spectroscopy. Morphologies indicated a network of aggregated nanotubes. The phase and composition analyses revealed that the lanthanide TNTs had anatase phases with Nd and/or Gd nanoparticles in the TNT lattice. The nanoparticles were uniformly deposited on the surface because of hydroxyl groups on the TNT surfaces, resulting in a very high loading density. The outer diameter and the length of the TNTs increased with doping. The mechanisms for the formation of multiwall TNTs are discussed.

Keywords

TiO₂ Nanotube, Lanthanide Doped, Hydrothermal, XRD, TEM, Raman Spectroscopy

1. Introduction

In the effort to improve the performance of fuel cells, dye-sensitized solar cells [1], photocatalysis [2] [3], and solar driven processes [4], the fabrication and electrocatalytic properties [5] [6] of metals deposited on nanomaterials have been intensely investigated. Nanoscale materials derived from titanium oxide (TiO₂) have been extensively investigated for various applications, including solar-cells and batteries [7] [8], hydrogen generation [9] [10] [11], drug delivery [12] [13], photolysis [14] [15], and electrocatalysis [16] [17]. Titanium nanotubes (TNT)

can be fabricated via the template-free hydrothermal method, which is simple, cost-effective, and environmentally friendly [18]. Furthermore, TiO_2 is a low-cost abundant resource known for its long-term stability, non-toxicity, and resistance to photo-corrosion.

Lanthanide ions have a special $4f^x5d^y$ electronic structure that enables them to form complexes with various compounds, such as TiO_2 , that can donate a pair of nonbonding electrons. Incorporation of lanthanide ions in a TiO_2 matrix could concentrate organic pollutants on the surface and thus enhance electrocatalytic and/or photocatalytic activity [19] [20] [21] [22]. The specific surface area of the nanoscale catalyst and its particle size are very important parameters that strongly affect its catalytic activity.

In recent years, researchers used many Lanthanide elements to modify the TiO_2 nanostructure (such as La, Pr, Nd, Eu, Gd, Tb, Ho, Er and Yb) by different method to improve its photocatalytic and others applications [20]-[31]. In particular, Lanthanide elements (such as La, Eu, Gd and Ce) doped TNTs were prepared by hydrothermal method using NaOH to improve its photocatalytic and photoelectrocatalytic activity for oxidation of organic dye and organic compounds [20] [23] [29] [31]. Doping methods include wetness impregnation method [20], electrochemical anodization [21] [22] [27] [30], precipitation methods [24], sol-gel method [19] [25], hydrothermal treatment [26], wetness impregnation method [20], ultrasonic hydrothermal [29], ion exchange [31].

Here, the hydrothermal method for TNT syntheses and an *in-situ* doping strategy was used to incorporate Nd and/or Gd nanoparticles in TNT scaffolds. The effects of these lanthanides on the structural, textural, and morphological properties of TiO_2 nanomaterials obtained after calcination of hydrogen titanate nanotubes were investigated. The synthesized TNT, Nd and/or Gd-TNT show different applications, such as great performance of electro-oxidation of hydrazine by electrosensory [32] and determination of pharmaceutical compound as Alendronate sodium [33].

2. Experimental

2.1. Synthesis of TNTs

TiO_2 powder [P25, (99.5%, 21 nm)] and potassium hydroxide (KOH) were purchased from Sigma-Aldrich, USA. HCl for acid washing was purchased from S.D. Fine Chemicals, India. TNTs were synthesized via the hydrothermal method, where a mixture of 0.5 g of TiO_2 powder was dispersed in 30 mL of concentrated aqueous KOH solution. The mixture was stirred for 30 min and then transferred into a Teflon-lined stainless-steel autoclave, where the hydrothermal treatment was performed for 24 h at 150°C [34]. The precipitate was collected and washed with deionized water and dilute HCl until the pH was 6.5. The TNTs were then dried for 10 h at 90°C , followed by annealing at 400°C for 2 h. In the case of doping, the molar ratio of titanium/dopant was 20:1 when $\text{Nd}(\text{NO}_3)_3$

and/or $\text{Gd}(\text{NO}_3)_3$ were added to the TiO_2 in the KOH solution. The hydrothermal and post-synthetic treatments were then performed as described above for the un-doped TNTs.

The samples were accordingly labelled TNTs, Nd-TNTs, Gd-TNTs, and Nd-Gd-TNTs. Finally, the TNTs became a white powder after drying and grinding. They were characterized with respect to surface morphology and crystallographic structure.

2.2. Apparatus

2.2.1. Morphology and Compound Composition

The morphological characterization of un-doped TNTs and doped TNTs prepped by hydrothermal method was achieved using Transmission Electron Microscope (TEM, JEOL JEM 1400, Japan) at 110 kV. The samples were prepared by dropping the ethanol solution of TNTs and Nd-TNTs, Gd-TNTs and Nd-Gd-TNTs catalysts on the Cu coating by carbon grids. The scanning electron microscopy (SEM) images were taken at different magnifications, without coating with conductive material, by (SEM, Super scan SS-550, Shimadzu, Japan). EDX was used to characterize the distribution of elements in un-doped TNTs and doped-TNTs synthesis by both hydrothermal using (EDX, Superscan SS-550, Shimadzu, Japan).

2.2.2. Crystalline Phase

Crystalline phases were acquired via X-ray diffraction (XRD, Shimadzu, XRD-7000, Japan) at 40 kV and 30 mA, using a Cu $K\alpha$ incident beam (0.154 nm). The TNT molecular structures were characterized with Fourier-transform infrared spectroscopy (FT-IR, Thermo Fisher Scientific Inc., Madison, WI, USA) over the range 400 - 4000 cm^{-1} . Raman spectroscopy was performed with 532-nm excitation on a Raman microscope (Sentarra, Bruker, USA) coupled to a Leica microscope (Olympus BX series, USA). The spectra were analyzed over the range 50 - 1200 cm^{-1} , with Rayleigh rejection via a 532 nm.

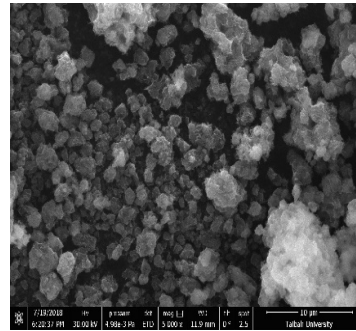
2.2.3. Surface Area and Porosity

Brunauer-Emmett-Teller specific surface areas (S_{BET}) of the TNTs were determined by a multipoint method using adsorption data for a relative pressure of 0.1515 (Micromeritics analyzer, Gemini VII, 2390 Surface Area and Porosity, USA).

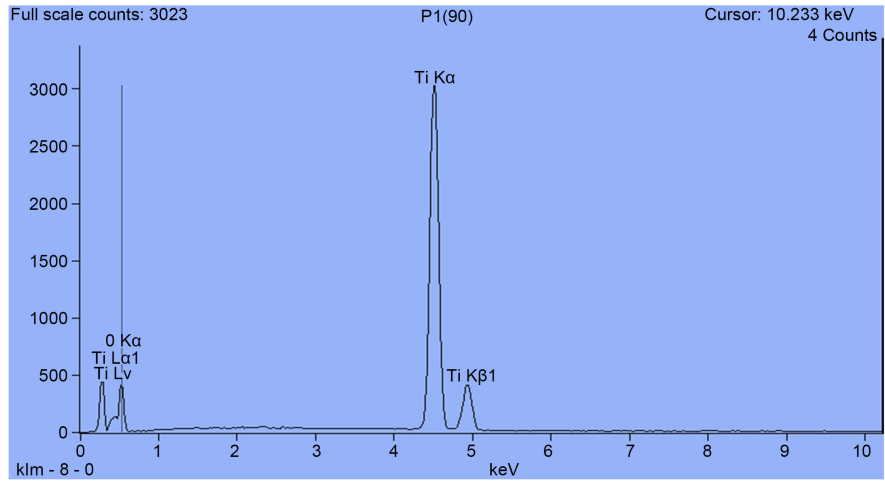
3. Results and Discussion

3.1. Nanotube Morphologies

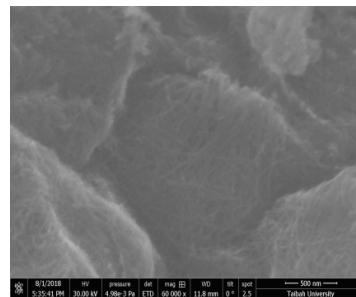
In **Figure 1(a)**, a spherical shape was observed for the P25 TiO_2 powder. The un-doped TNTs samples calcined at 400 °C (**Figure 1(b)**) were aggregated thread networks. **Figures 1(c)-(e)** are SEM images of in situ lanthanide-doped TNTs calcined at 400 °C. These nanotubes were heavily aggregated.



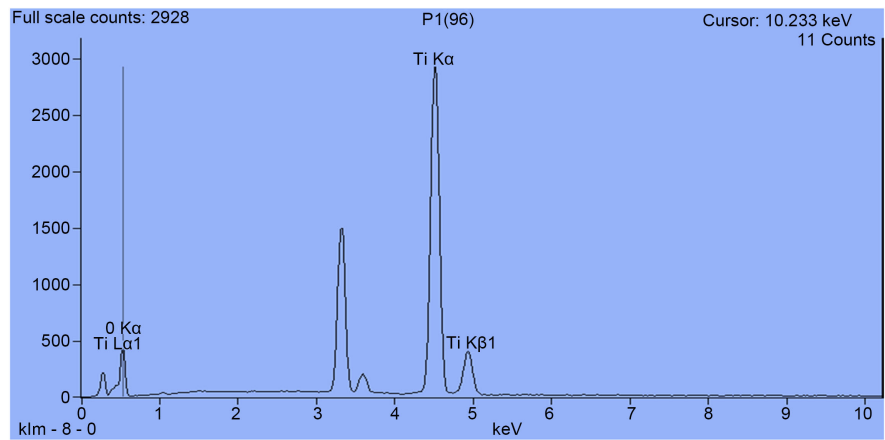
(a)



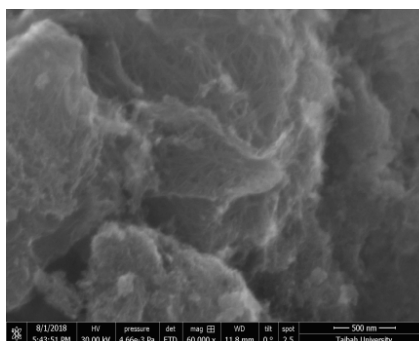
(a')



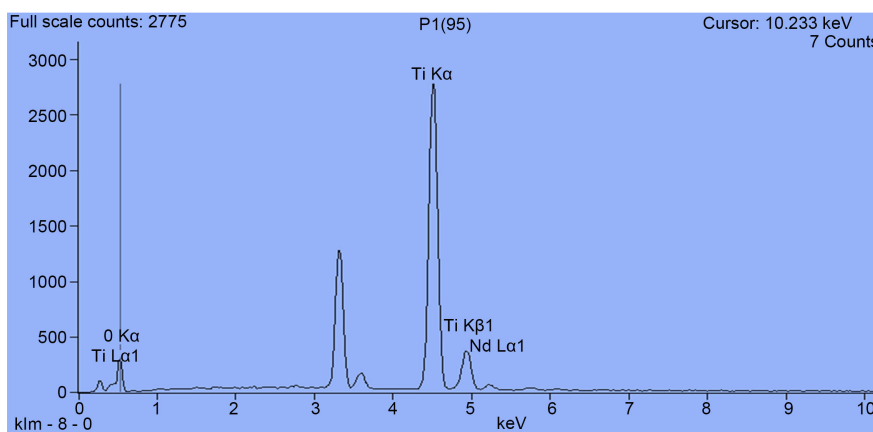
(b)



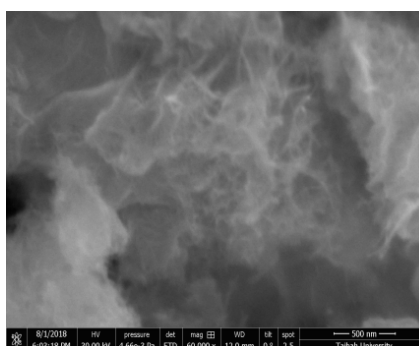
(b')



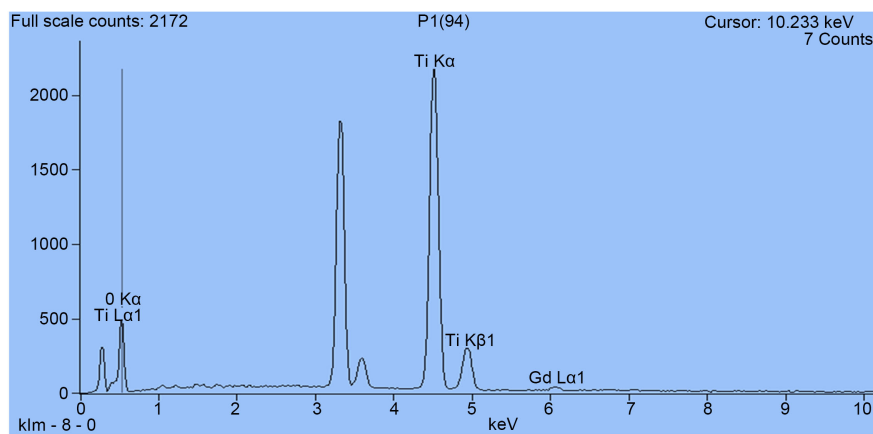
(c)



(c')



(d)



(d')

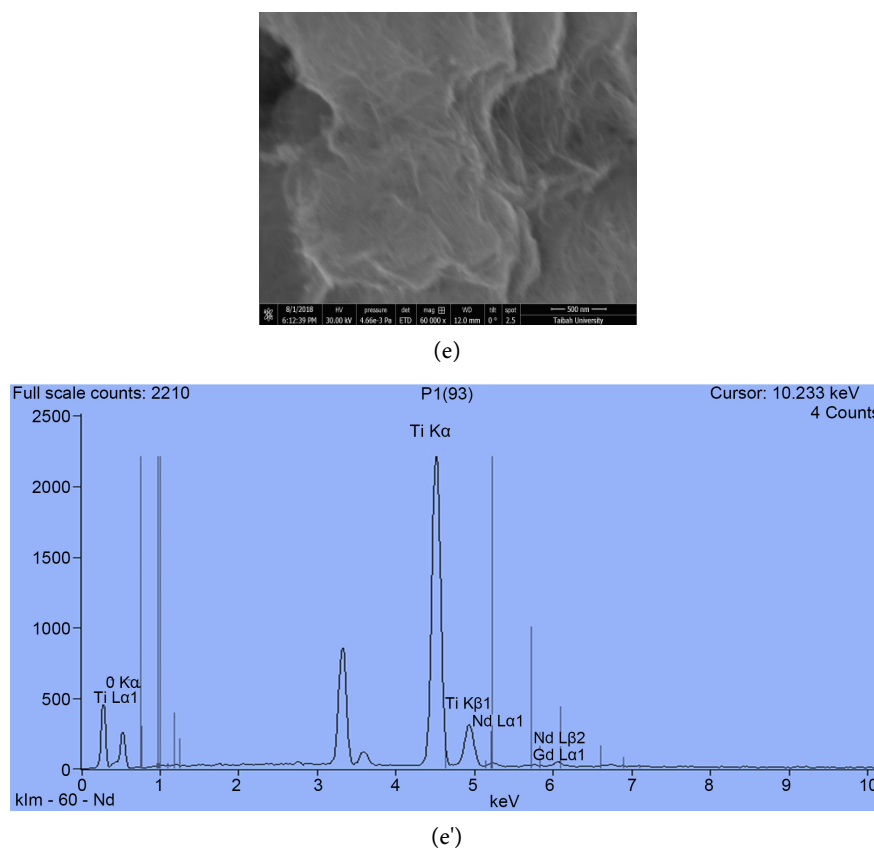


Figure 1. SEM images of ((a), (b), (c), (d)) P25 TiO_2 Powder, undoped TNTs, Nd-TNTs, Gd-TNTs, and Nd-Gd-TNTs; ((a'), (b'), (c'), (d')) EDX of undoped TNTs, Nd-TNTs, Gd-TNTs, and Nd-Gd-TNTs respectively.

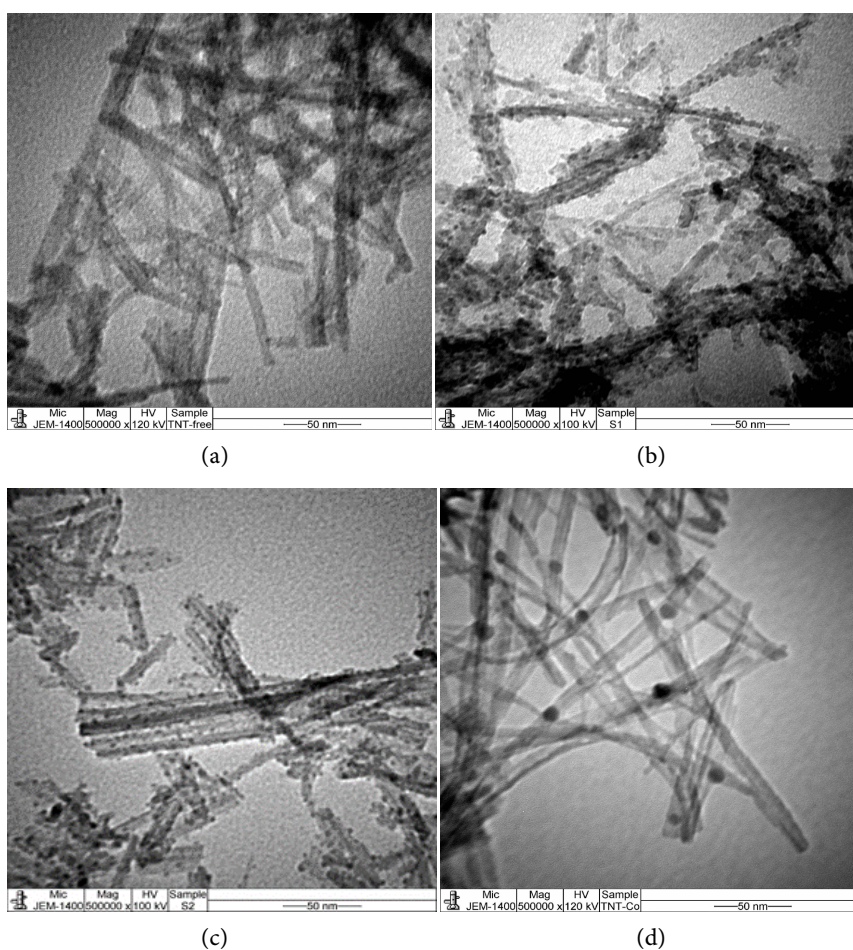
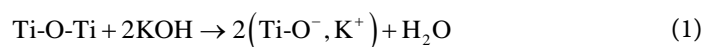
The approximate compositions of P25 and the un-doped and doped TNTs were calculated from EDX spectra (**Figures 1(a')-(e')**). Uniform distributions of Nd and Gd nanoparticles were observed in the TNT lattice. As shown in **Table 1**, the EDX patterns had nearly 3.37% Nd loading in the Nd-TNTs, 2.92% Gd loading in the Gd-TNTs, and 9.42% Nd and 3.48% Gd loading in the Nd-Gd TNTs.

Figure 2(a) displays a TEM image of un-doped TNTs that were aggregated and formed a large composite fiber-like structure, in agreement with the SEM images. The nanotubes were uniform and hollow, with multiwall sheets. The outer diameters were 6.5 - 10.6 nm, and the lengths were 51 nm. The open-ended fiber-like structures of uniform, multiwall, straight tubes are shown in **Figures 2(b)-(d)**. The well-dispersed Nd and/or Gd oxide nanoparticles in the TNT lattice also can be seen. The outer diameter of the Nd-TNT multi-layered sheets was in the range of 5.7 - 8 nm, that for the Gd-TNTs was in the range of 6.1 - 14.2 nm, and that for the Nd-Gd-TNTs was in range of 9.4 nm - 14.2 nm.

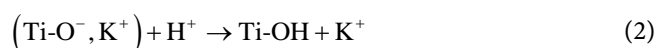
The mechanisms for multiwall TNT formation by hydrothermal treatment could be understood as follows. When the Ti precursors were mixed with the concentrated aqueous KOH at high temperature and pressure, Ti-O-K bonds formed on the TiO_2 surface instead of "Ti-O-Ti" bonds because of the concentrated KOH [35]. These salts formed nanosheets by:

Table 1. EDX analysis of P25 TiO₂, undoped TNTs, Lanthanides-doped TNTs.

| Spectrum Label | O% | Ti% | Nd% | Gd% | Total |
|----------------------|-------|-------|------|------|--------|
| P25 TiO ₂ | 46.56 | 53.44 | --- | --- | 100.00 |
| Undoped TNTs | 49.07 | 50.93 | --- | --- | 100.00 |
| Nd-TNTs | 40.97 | 55.66 | 3.37 | --- | 100.00 |
| Gd-TNTs | 53.55 | 43.53 | --- | 2.92 | 100.00 |
| Nd-Gd-TNTs | 41.72 | 45.38 | 9.42 | 3.48 | 100.00 |

**Figure 2.** TEM images of (a) undoped TNTs, (b) Nd-TNTs, (c) Gd-TNTs, and (d) Nd-Gd-TNTs.

Then, when the samples were treated with deionized water, the Ti-O-K bonds in the (K₂Ti_nO_{2n+1}) multilayer crystal gradually converted into Ti-OH bonds in sheets of hydroxyl titanate (H₂Ti_nO_{2n+1}). After the sample was treated by dilute acid, Ti-O-Ti bonds [36] or Ti-O-H...O-Ti hydrogen bonds were generated according to:



These bonds significantly decreased the Ti-O length, thus leading to the folding of the sheets into a multiwall tube structure, as shown in **Figure 3**.

3.2. Crystalline Phase Characterization

The crystal phases of P25 TiO₂ and un-doped and doped TNTs are shown in **Figure 4**. The XRD patterns in **Figure 4(e)** for P25 TiO₂ indicated anatase phase planes with main peaks located at 25.3°, 37.8°, 48.0°, and 55.1°, respectively corresponding to the (101), (004), (200), and (211) planes, in good convention with the standard spectrum [Joint Committee on Powder Diffraction Standards (JCPDS), card no.: 21-1272] [37]. The XRD patterns of un-doped TNTs are shown in **Figure 4(a)**, where diffraction peaks were observed at 24.90°, 48.10°, and 55.91°, corresponding to the (101), (200), and (211) crystal planes of anatase TiO₂, respectively. These data suggested an anatase phase of TiO₂ (JCPDS, card no.: (00-021-1272)). However, these XRD patterns were typical of layered materials, suggesting that the TNTs were multi-walled [38]. The diffraction peak with a low 2θ value of 11.5° (020) corresponded to the TNT interlayer distance. An intense peak at 2θ = 48.10° was due to the edge-sharing TiO₆ octahedral [18].

The diffraction planes of anatase in **Figure 4(b)** were sharp, indicating good crystallization of Nd-TNTs, and the other four peaks had the same positions of 22.70°, 32.45°, 46.65°, and 57.28° assigned to (002), (200), (220), and (132) reflections, indicating well-embedded incorporation of Nd ions in the TiO₂ lattice as NdTiO₃ (JCPDS, card no.: (00-029-0922)). Meanwhile, in **Figure 4(c)**, there were characteristic peaks of Gd observed at 29.28°, 31.64°, 47.89°, and 58.72° assigned to (201), (040), (341), and (611) planes, respectively, indicating Gd ions in the form of Gd₂TiO₅ (JCPDS, card no.: (00-021-0342)). The XRD patterns recorded for co-doped Nd-Gd-TNTs are shown in **Figure 4(d)**. They revealed anatase phases, with new peaks for Nd and/or Gd.

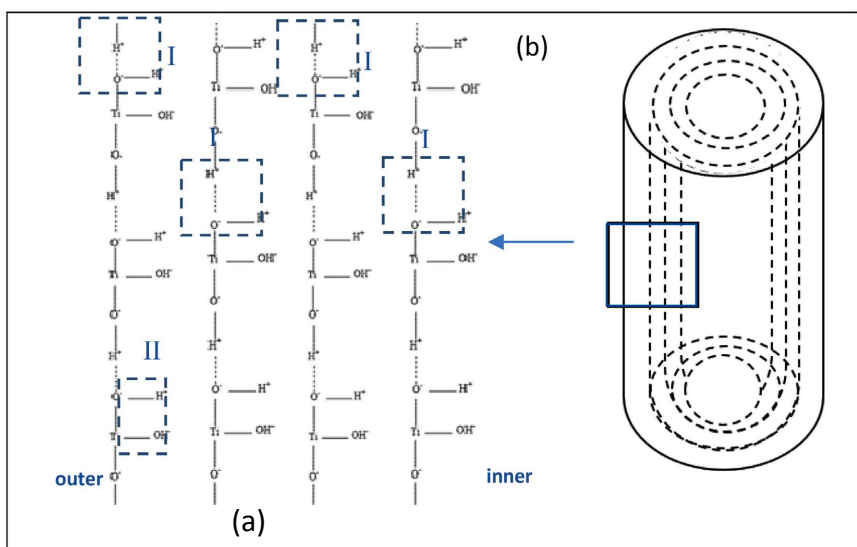


Figure 3. Hydroxyl titanate nanosheets ($H_2Ti_nO_{2n+1}$), (a) mechanism for breaking of $Na_2Ti_2O_4(OH)_2$, (b) rolling up the nanosheet to nanotubes.

previous work. The E_g peak was attributed mainly to the symmetric stretching vibration of O-Ti-O linkages in TiO_2 , the symmetric bending vibration of O-Ti-O was confirmed by B_{1g} , and the A_{1g} peak was associated with the antisymmetric bending vibration of the O-Ti-O linkage [33]. The same strong peaks observed in **Figures 5(b)-(d)** for the doped TNTs had anatase characteristics, indicating that Nd and Gd may be introduced into the lattice or interstitial sites of TiO_2 .

Table 2. Particle sizes, BET surface area, pore volume and pore size of synthesis TNTs

| Sample | Average Crystalline Size (nm) | BET Specific Surface Area, S_{BET} (m^2/g) | Pore Volume ($cm^3 \cdot g^{-1}$) | Pore Size (\AA) |
|--------------------------------|-------------------------------|--|-------------------------------------|----------------------------|
| P25 TiO_2 | 7.31 | 58.90 | 0.047 | 16.9 |
| Undoped TNTs | 7.06 | 160 | 0.064 | 16.05 |
| Nd-TNTs (Ti:Nd = 20:1) | 7.93 | 175 | 0.068 | 16.07 |
| Gd-TNTs (Ti:Gd = 20:1) | 7.91 | 207 | 0.083 | 16.06 |
| Nd-Gd-TNTs (Ti:Nd + Gd = 20:1) | 6.35 | 231 | 0.073 | 16.11 |

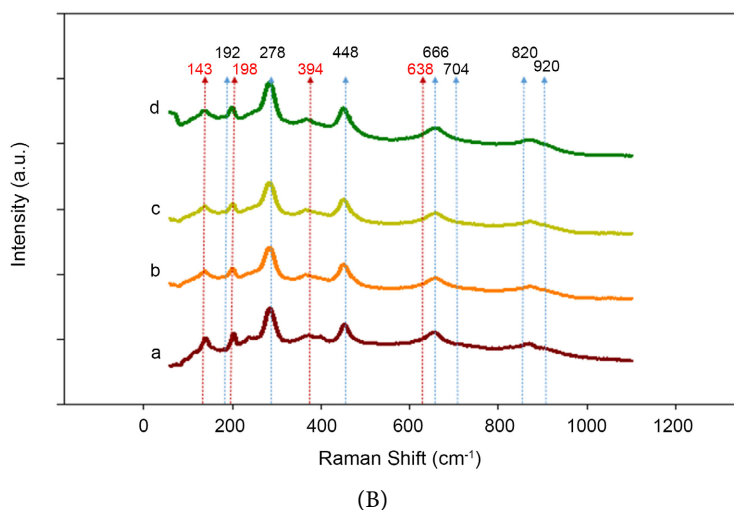
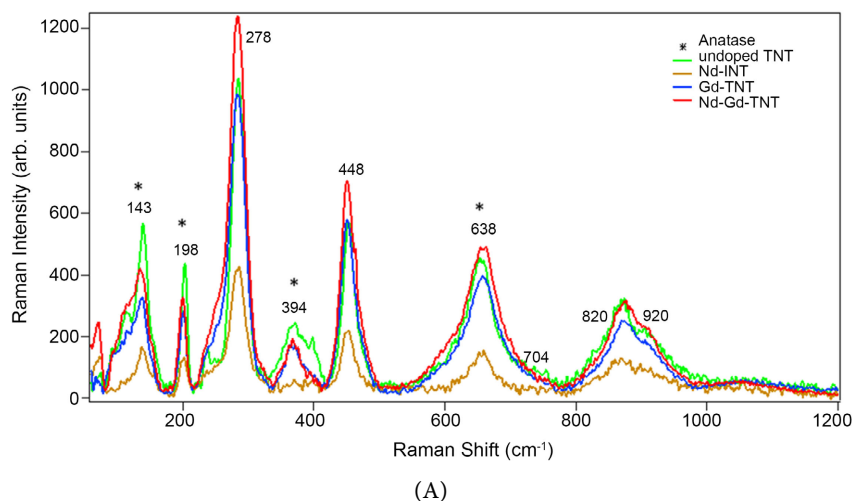
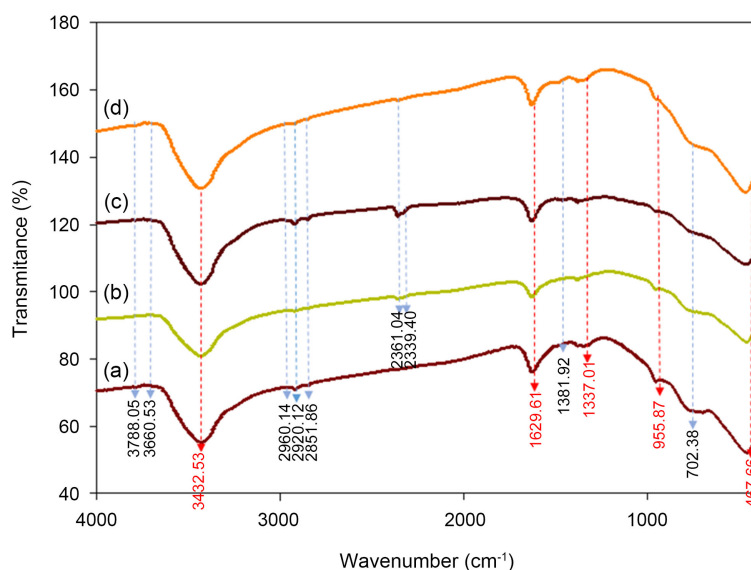


Figure 5. Raman spectra of (a) undoped TNTs, (b) Nd-TNTs, (c) Gd-TNTs, and (d) Nd-Gd-TNTs.

Table 3. Raman shift of vibration mode for different catalyst supports.

| Raman shift (cm ⁻¹) | Vibration | Ref. |
|---------------------------------|---|-----------|
| 143 | symmetric stretching vibration of O-Ti-O in anatase form | [45] |
| 198 | symmetric stretching vibration of Ti-O-Ti in anatase form | [43] [46] |
| 278 | symmetric stretching vibration of Ti-O-K bonds | [50] |
| 394 | symmetric binding vibration of Ti-O-Ti in anatase form | [42] |
| 448 | symmetric stretching vibration of Ti-O-Ti crystal phonons | [46] |
| 638 | symmetric stretching vibration of Ti-O-Ti crystal phonons | [46] |
| 704 | symmetric stretching vibration of covalent Ti-O-H bonds | [46] |
| 820 | symmetric stretching vibration of covalent Ti-O-H bonds | [47] |
| 920 | surface Ti-O-K vibrations | [47] |

**Figure 6.** FT-IR spectroscopy of (a) TNTs, (b) Nd-TNTs, (c) Gd-TNTs, and (d) Nd-Gd-TNTs.

The band at 278 cm⁻¹ was assigned to the stretching vibration of Ti-O-K bonds [45], while the bands at 198 cm⁻¹ and 394 cm⁻¹ corresponded to anatase Ti-O-Ti [43]. The bands at 448 cm⁻¹ were related to Ti-O-Ti crystal phonons [46]. The bands at 704 cm⁻¹ and 820 cm⁻¹ corresponded to covalent Ti-O-H bonds, and the band at 920 cm⁻¹ was assigned to surface Ti-O-K vibrations [47] (Table 3).

3.5. Fourier-Transform Infrared Spectra

In FT-IR spectra of TNTs (Figure 6), three absorption bands centered at 3432.53 cm⁻¹, 1627.87 cm⁻¹, and 955.87 cm⁻¹ were assigned to stretching vibrations of OH groups, and bending vibrations of H-O-H and Ti-O bonds, respectively [48]. The strong intensity of the hydroxyl groups at 3432.53 cm⁻¹ indicated a large concentration on the un-doped TNT surface. This helped conjunction and

dispersion of the metal particles on the TNT walls [49] [50] [51] [52], and also captured photoexcited electrons and holes to produce reactive oxygen species for photocatalysis of organic wastewater. The peaks at 1627.87 cm^{-1} and 1349.52 cm^{-1} corresponded to undoped TNTs [50]. The unchanged intensity of the 3432.53 cm^{-1} band may have been due to the interactions of doped Nd and Gd in the TiO_2 lattice that did not replace -OH groups on the oxide surface. Moreover, the band at $446 \pm 4\text{ cm}^{-1}$ corresponded to the asymmetric vibration of M-Ti-O (M = K, Nd, Gd) groups inside the lattice [27]. In all cases, the FT-IR spectra kept their general features, which confirmed that the metals and the TNTs had strong chemical interactions, consistent with the XRD and Raman structural analyses.

4. Conclusion

The fabrication of un-doped TNTs and Nd- and/or Gd-doped TNTs was demonstrated via the environmentally-friendly hydrothermal treatment of TiO_2 P25 in a concentrated KOH solution and *in-situ* doping. The morphologies of all the samples from SEM, EDX, and TEM data confirmed that the structure of the multiwall nanotubes was fiber-like with a uniform distribution of Nd and/or Gd ions in the TNT lattice. The nanotubes were in an anatase phase, even after doping with Nd and/or Gd oxide. The mixed Nd-Gd-TNTs had smaller grain sizes because the dopant cations were located at the grain boundary, thus hindering grain growth. It was found that the specific surface area of the nanotubes ($50 - 210\text{ m}^2/\text{g}$) considerably exceeded that of the initial TiO_2 ($12\text{ m}^2/\text{g}$).

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Contribution

Emran and Alanazi carried out the experiments. Emran developed the theory and performed the computations with Alanazi. Emran encouraged Alanazi to investigate the preparation of TNTA and supervised the findings of this work. Both authors discussed the results and contributed to the final manuscript.

Conflicts of Interest

The authors declare no conflicts of interest regarding the publication of this paper.

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